Accepted version on Author's Personal Website: C. R. Koch

Article Name with DOI link to Final Published Version complete citation:

Hadi Nazaripoor, Charles R. Koch, and Subir Bhattacharjee. Electrical perturbations of ultrathin bilayers: Role of ionic conductive layer. *Langmuir*, 30(49):14734–14744, 2014. doi: 10.1021/la503839x. PMID: 25419880

See also:

https://sites.ualberta.ca/~ckoch/open_access/Hadi_langmuir_2014.pdf



As per publisher copyright is $\bigcirc 2014$



(c) (\$) (E)

This work is licensed under a

Creative Commons Attribution-NonCommercial-NoDerivatives 4.0 International License.





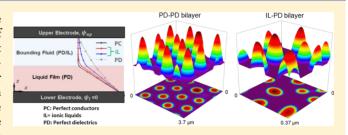
Article

pubs.acs.org/Langmuir

¹ Electrical Perturbations of Ultrathin Bilayers: Role of Ionic ² Conductive Layer

- ³ Hadi Nazaripoor, [†] Charles R. Koch,*,[†] and Subir Bhattacharjee[‡]
- 4 [†]Department of Mechanical Engineering, University of Alberta, Edmonton, Alberta T6G 2R3, Canada
- s [‡]Water Planet Engineering, 721 Glasgow Ave, Unit D, Inglewood, California 90301, United States

ABSTRACT: The effect of electrostatic forces on the dynamics, morphological evolution, and drainage time of ultrathin liquid bilayers (<100 nm) are investigated for perfect dielectric—perfect dielectric (PD-PD) and ionic liquid—perfect dielectric (IL-PD) bilayers. The weakly nonlinear "thin film" equation is solved numerically to obtain spatiotemporal evolution of the liquid—liquid interface responses to transverse electric field. In order to predict the electrostatic component of conjoining/disjoining pressure



acting on the interface for IL-PD bilayers, an analytical model is developed using the nonlinear Poisson-Boltzmann equation. It is found that IL-PD bilayers with electric permittivity ratio of layers (lower to top), ε_r , greater than one remain stable under an applied electric field. An extensive numerical study is carried out to generate a map based on ε_r and the initial mean thickness of the lower layer. This map is used to predict the formation of various structures on PD-PD bilayer interface and provides a baseline for unstable IL-PD bilayers. The use of an ionic liquid (IL) layer is found to reduce the size of the structures, but results in polydispersed and disordered pillars spread over the domain. The numerical predictions follow similar trend of experimental observation of Lau and Russel. (Lau, C. Y.; Russel, W. B. Fundamental Limitations on Ordered Electrohydrodynamic Patterning; *Macromolecules* **2011**, *44*, 7746–7751).

INTRODUCTION

6

8

9

10

11

12

13

14

15

16 17

18

19

20

21

24 Thin films and surfaces are commonly applied to decorative or 25 protective coatings, an intervening media in liquid-liquid ²⁶ emulsions, ^{2,3} and occur as a lubricant in the cornea of the eye. ⁴ In 27 technological processes where the stability of thin films is 28 important, numerous theoretical and experimental studies have 29 been conducted to understand the factors that lead to the rupture 30 of the thin films. 1-13 Thin films are unstable when the 31 wavelength of fluctuations is larger than a critical value.⁵ 32 Perturbations can lead to a diverse range of dynamics ranging 33 from stable films to ruptured films showing dry spots/or holes. 34 This behavior is attributed to the intermolecular interactions 35 which are typically not significant in bulk fluid dynamics. 6 A 36 pressure balance at the free interface is used to determine the 37 shape between the layers. Thin film instability can be triggered 38 naturally^{9,14,15} (i.e., intermolecular forces) and by externally 39 applied forces^{2,13,16–23} (i.e., mechanical, thermal, and electric 40 forces). Films with thicknesses of 100 nm or less are called 41 ultrathin films and can be either stable or unstable depending on 42 the sign of the second derivative of the interaction potential with 43 respect to film thickness.⁵ Natural intermolecular interactions 44 (such as van der Waals interactions, apolar, 5,8,24 and polar 45 interactions⁹) determine the initial state of a thin film on a 46 substrate. The type and magnitude of the interactions depend on 47 properties of system materials.^{25,26}

Applying an electric field can also cause film motion due to 49 creation of a stress at the film interface. Electrically induced 50 instability of thin films has received extensive attention as a tool for generating novel submicrometer structures, ^{16,19,20,27–31,31–45} 51 with applications in the fabrication of microfluidic devices, 52 biosensors, fuel and solar cells, and optoelectronic devices. 53 Among several micro and nanofabrication methods (e.g., self-54 assembly, soft lithography, optical lithography, nanoimprinting, 55 etc.), electrohydrodynamic lithography (EHL) is a promising 56 technique to create micron and submicron patterns at the thin 57 liquid/molten polymer film interface. This is due to the capability 58 of EHL in generating complex structures rapidly and 59 economically.³² Raising the electric field above a critical value 60 results in crease formation in the thin film and a further increase 61 in the electric field can cause crater formation on the polymer 62 interface. ^{19,43}

Electrohydrodynamic (EHD) pattern formation is an energy 64 driven process³⁰ where the dynamics of structures (i.e., pillars, 65 holes, and bicontinuous pattern formation) form to minimize the 66 main energy component, the electrostatic energy. From a 67 thermodynamic point of view, the film evolves to lower the free 68 energy of system^{29,30} such that at steady state the lowest 69 interfacial energy determines the final structure.

Initial stages of film evolution can be described by linear 71 stability (LS) analysis, whereas further stages show nonlinear 72 behavior. Feature size in the EHD process is characterized 73 by the fastest growing wavelength of instabilities, $\lambda_{\rm max}$. The 74

Received: July 25, 2014 Revised: November 22, 2014



75 center-to-center distance of pillars in (typically a hexagonal pillar 76 pattern) can be predicted by LS analysis. 16,27,29,32,39,41 In LS 77 analysis, λ_{max} is a function of electric permittivity of layers ε , electrode distance d_r , initial mean film thickness h_0 , applied voltage V_r , and interfacial tension γ . Bilayers with $\varepsilon_r > 1$ have been widely studied both numerically 29,34,41,47,48 and experimentally; 16,27,31,33,35,39 bilayers with $\varepsilon_r < 1$ have received 82 less attention despite their potential for producing smaller 83 features as demonstrated by experiments. 27,39

Dynamics and spatiotemporal evolution of perfect dielectric (PD) and leaky dielectric (LD) bilayers which are bounded with 86 air are numerically modeled for both flat and patterned 87 electrodes. 38,44 PD materials, with no free charge, polarize 88 under the electric field, but in LD materials, with infinitesimal 89 amount of charges, charges move and accumulate on the 90 electrode surface and the interface. 49 Significant decrease in the 91 length scale of structures is observed due to the presence of free 92 charges at the interface. 44,50 It is experimentally shown that the 93 use of an ionic liquid (IL) bounding layer results in 94 submicrometer-sized features even with the use of flat electro-95 des. ³⁹ Ionic liquids can be ionic salt like materials, considered as 96 high temperature ionic liquids, or low temperature ionic liquids ³⁹ which are liquid below 100 °C and electrolytes (aqueous or 98 nonaqueous).51

An IL-PD interface which is subject to an applied electric field 100 is present in many microfluidic system applications. Free ions present in the ionic liquids and under the applied electric field 102 redistribute and accumulate on the charged surfaces. The diffuse 103 layer of ions is called the electric double layer (EDL). The EDL structure is essential to obtain the electromechanical behavior of IL-PD interface. 51,52 This diffuse layer is not present in perfect dielectrics and leaky dielectrics. Another difference is the electroneutrality condition of bulk in the leaky dielectric case.

Dynamics of the free ions within IL material is governed by the 109 Poisson-Nernst-Planck equation which leads to the Boltzmann 110 distribution for an equilibrium condition⁵¹ where the charge relaxation time is much smaller than the process time.²⁶ It is observed that IL layers subjected to a strong electric field (over O(108) V/m) behave like a perfect conductor. 52 Since the applied voltage and the upper layer thickness in the EHD atterning process result in the IL layer experiencing both strong and weak electric field conditions, a comprehensive electrostatic model is needed to predict both perfect conducting and EDL 117 behavior of the IL layer.

An analytical model based on the coupled nonlinear Poisson-120 Boltzmann and Laplace equations to obtain the electric potential 121 distribution within layers and consequently the electrostatic 122 pressure acting on the interface is developed. Then, a 123 comprehensive numerical investigation is performed to inves-124 tigate the morphological behavior of bilayers. Layers with an electric permittivity ratio of less than one $(\varepsilon_r < 1)$ under a 126 homogeneous electric field are the focus of this investigation. The morphology of the film and its drainage time under the effects of filling ratio and electric permittivity ratio are studied. Finally, the perfect dielectric (PD) upper layer is replaced with an IL layer to investigate spatiotemporal behavior of the bilayer. Analytical calculations are performed and integrated into the 132 numerical simulation to obtain the interface potential and to 133 define an electrostatic component of conjoining/disjoining 134 pressure acting on the interface.

PROBLEM FORMULATION

A schematic of the thin film sandwiched between upper and 136 lower electrodes is shown in Figure 1a. Thin film evolution is 137 fl

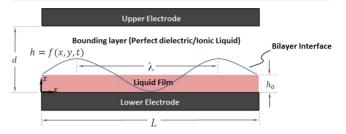


Figure 1. (a) Schematic view of the thin film sandwiched between electrodes. The z-position of the free interface is described by the function h = f(x, y, t) with the lateral coordinates x, y, and time t. The amplitude of the fluctuation with wavelength is not to scale and is magnified. d = 100 nm.

described using mass and momentum conservation for both the 138 thin film and the bounding fluid.

$$\nabla. \ \vec{u}_i = 0 \tag{1}$$

$$\rho_{i} \left(\frac{\partial \vec{u}_{i}}{\partial t} + (\vec{u}_{i}. \nabla \vec{u}_{i}) \right) \\
= -\nabla P_{i} + \nabla. \left[\mu_{i} (\nabla \vec{u}_{i} + (\nabla \vec{u}_{i})^{T}) \right] + \vec{f}_{e} \tag{2}$$

$$= -\nabla P_i + \nabla \cdot \left[\mu_i (\nabla \vec{u}_i + (\nabla \vec{u}_i)^T) \right] + f_e \tag{2}$$

These equations are solved with the following boundary 142 conditions: no slip condition on the walls ($\vec{u}_i = 0$ at z = 0 and 143 z = d), no penetration (two media are immiscible: $\vec{u}_{\text{relative}} = 0$) and 144 stress balance (normal: $\vec{n} \cdot [\overline{\sigma}_1 \cdot \vec{n} - \overline{\sigma}_2 \cdot \vec{n}] = \kappa \gamma + \vec{f}_e \cdot \vec{n}$ and tangent: \vec{t}_i . 145 $[\overline{\sigma}_1.\vec{n} - \overline{\sigma}_2.\vec{n}] = \vec{f}_e.\vec{t}_i)$ at the interface (z = h(x, y, t)), where $\overline{\sigma} = -P\overline{I}$ 146 $+ \mu_i (\nabla \vec{u}_i + (\nabla \vec{u}_i)^T)$ and $\vec{f}_e = -\nabla \phi$ is for external body force, ϕ is 147 conjoining pressure, μ_i is the dynamic viscosity, and subscript i 148 denotes the fluid phase (here 1 and 2 for the thin film and the 149 bounding media, respectively). The normal and tangent vectors 150 of the interface are \vec{n} and \vec{t}_i , respectively. ⁵³ The surface tension, γ , 151 is assumed to be constant and κ is the mean interfacial curvature 152 of the film interface. 53

The liquid layers are assumed Newtonian, incompressible, and 154 have an electric permittivity of $\varepsilon_1(\varepsilon_2)$ for lower (upper) layer. 155 The film is assumed sufficiently thick that the continuum 156 assumption is valid, but thin enough that gravity effects can be 157 neglected. The system is assumed isothermal so energy 158 conservation is not needed. A kinematic boundary condition is 159 used to relate interfacial velocity component to the film thickness 160

$$w = \frac{\partial h}{\partial t} + u \frac{\partial h}{\partial x} + v \frac{\partial h}{\partial y} atz = h(x, y, t)$$
(3) ₁₆₂

In the EHL process, an applied electric field destabilizes the 163 interface of the liquid film and bounding fluid, leading to the 164 evolution of patterns with height of h = f(x, y, t). The wavelengths 165 of the growing instabilities, λ , are much larger than the initial film 166 thickness (at initial stages), so a "long-wave approximation" 167 used to simplify the governing equations. Inertial effects are also 168 negligible because the thickness of the lower layer remains less 169 than 100-200 nm during the film dynamics, which leads to 170 Reynolds number which is much smaller than one (Re < 1). To 171 reduce the computational effort, the polymer film (lower layer) is 172 assumed much more viscous than the bounding fluid (upper 173

174 layer). Applying these assumptions to the governing equations 175 and boundary conditions results in the following "thin film" 176 equation: 12,24,29,41,54

$$3\mu h_t + \frac{\partial}{\partial x} \left[h^3 \frac{\partial \psi}{\partial x} \right] + \frac{\partial}{\partial y} \left[h^3 \frac{\partial \psi}{\partial y} \right] = 0$$

$$\psi = \gamma [h_{xx} + h_{yy}] - \phi \tag{4}$$

178 which is a long-wave limit description of spatiotemporal 179 evolution of film interface, h(x, y, t). In ψ , $\gamma[h_{xx} + h_{yy}]$ is the 180 surface tension force, while ϕ represents the excess intermo-181 lecular forces, which is a conjoining pressure as mentioned 182 previously. Laplace pressure (the first term in ψ) is a result of 183 local curvature of the free interface and interfacial tension. By 184 minimizing the interface area of the film and the free energy of 185 system, this term damps fluctuations. The viscous term, $3\mu h_{\nu}$ 186 also damps fluctuations in the fluid film. Increasing the films 187 viscosity increases film evolution times. S

Interaction Potentials. Conjoining pressure, ϕ , (force 189 acting on the film interface per unit area) is defined as the 190 gradient of excess intermolecular interactions, $\phi = \nabla(\Delta G)$. In 191 EHD, the electric field induces a pressure at the film interface which is added to the natural interactions to generate excess 193 intermolecular interactions. The conjoining pressure, ϕ , is a 194 summation of these interaction potentials: van der Waals, 195 electrostatic, and Born repulsive interaction potentials and is 196 given as

$$\phi = \phi_{\text{vdW}} + \phi_{\text{Br}} + \phi_{\text{EL}} \tag{5}$$

198 The van der Waals interaction is the summation of Keeson, 199 Debye, and London dispersion forces. ^{25,26} This interaction is 200 defined as $\phi_{\rm vdW_L} = A_{\rm L}/6\pi h^3$ and $\phi_{\rm vdW_U} = -A_{\rm U}/6\pi (d-h)^3$ for the 201 lower and upper electrodes, respectively. $A_{\rm L}$ and $A_{\rm U}$ are effective 202 Hamaker constants for the electrodes which depend on the 203 materials used for electrodes and fluid layers. ^{25,26}

The van der Waals interaction are singular as $h \to 0$ and $h \to d$. To avoid nonphysical penetration of liquid to solid phase, in case of film rupture and touching, a cutoff distance, l_0 , is defined for which a short-range repulsive force, called Born repulsion, acts on the film interface. This is used to maintain a minimum equilibrium liquid thickness on both electrodes and is defined as $\phi_{\rm Br_L} = -8B_{\rm L}/h^9$ and $\phi_{\rm Br_U} = 8B_{\rm U}/(d-h)^9$ for the lower and upper electrodes, respectively. Coefficients $B_{\rm L}$ and $B_{\rm U}$ are found by setting the net conjoining pressure equal to zero at $h = l_0$ and $h = d - l_0$ for lower and upper surfaces. The most significant component of the conjoining pressure is the electrostatic pressure developed due to Maxwell stress to a significant h = 10 pressure developed due to Maxwell stress to an h = 10 pressure developed due to Maxwell stress to an h = 10 pressure developed due to Maxwell stress to an h = 10 pressure developed due to Maxwell stress to an h = 10 and h = 10 pressure developed due to Maxwell stress to an h = 10 pressure developed due to Maxwell stress to an h = 10 pressure developed due to Maxwell stress to an h = 10 pressure developed due to Maxwell stress to an h = 10 pressure developed due to Maxwell stress to an h = 10 pressure developed due to Maxwell stress to an h = 10 pressure developed due to h = 11 pressure developed due to h = 12 pressure developed due to h = 12 pressure developed due to h = 13 pressure developed due to h = 14 pressure developed due to

$$\phi_{\rm EL} = \frac{1}{2} \varepsilon_0 \left[\varepsilon_1 \left(\frac{\partial \psi_1}{\partial z} |_{z=h} \right)^2 - \varepsilon_2 \left(\frac{\partial \psi_2}{\partial z} |_{z=h} \right)^2 \right]$$
 (6)

217 where ε_1 (ε_2) is the relative electric permittivity of lower (top) 218 layer and ε_0 is free space electric permittivity. For the air—219 polymer bilayers (i.e., PD—PD materials with no free charge), the 220 Laplace equation governs the electric potential in the 221 layers. ^{16,29,41}

$$\varepsilon_i \varepsilon_0 \nabla^2 \psi_i = 0 \tag{7}$$

For the IL layer case, with *N* numbers of species in the lectrolyte, the Poisson equation is used to obtain electric potential in the IL layer.

$$\varepsilon_i \varepsilon_0 \nabla^2 \psi_i = -\rho_f \tag{8}$$

The boundary conditions are the applied potential on the 227 upper electrode, grounded for the lower electrode, and the 228 electric displacement and potential continuity at the bilayer 229 interface. In eq 8, ρ_f accounts for the free charge distribution 230 within the IL layer which is related to the ionic number 231 concentration as $\rho_f = \sum_{i=1}^N \overline{z_i} e n_i$. Free ions move and stay close to 232 the charged surfaces to form a Debye layer. The dynamics of 233 the free ions become insignificant when using a Boltzmann 234 distribution. The Boltzmann distribution must satisfy the 235 electroneutrality condition within the IL layer and is derived 236 from ion conservation equation in an equilibrium condition

$$n_{i} = n_{i\infty} \exp \left[-\frac{\overline{z}_{i} e(\psi - \psi_{\text{ref}})}{k_{\text{B}} T} \right]$$
(9) 238

where $\psi_{\rm ref}$ and $n_{i\infty}$ are reference potential and bulk ion number 239 concentration, respectively. The reference potential $\psi_{\rm ref}$ is set as 240 the applied voltage. For the ith species in the electrolyte, \overline{z}_i is the 241 valence, e is the magnitude of electron charge, $k_{\rm B}$ is the 242 Boltzmann constant, and T is the temperature in K. Bulk ion 243 number concentration depends on the molarity of electrolyte, n_{∞} 244 = $1000N_AM$, for which $N_{\rm A}$ is the Avogadro number and M is the 245 electrolyte molar concentration. The nondimensional electric 246 potential and length are defined as $\Psi = ez\overline{\psi}/k_{\rm B}T$ and Z = z/(d-247h), respectively. The nondimensional Laplace and nonlinear 248 Poisson-Boltzmann equations in the long wave limit are

$$\frac{d^2\Psi_1}{dZ^2} = 0 \tag{10}_{250}$$

$$\frac{\mathrm{d}^2 \Psi_2}{\mathrm{d}Z^2} = (\kappa (d-h))^2 \sinh(\Psi_i - \Psi_{\text{ref}}) \tag{11}$$

These are solved to find electric potential distribution across the 252 PD layer, $0 \le Z \le (h/(d-h))$ 253

$$\Psi_1 = \left(\frac{d-h}{h}\right) \Psi_s Z \tag{12}$$

and the IL layer, $(h/(d-h)) \le Z \le (d/(d-h))$

$$\Psi_{2} = \Psi_{\rm up} + 2 \ln \left[\frac{1 - \exp\left(\kappa (d - h) \left(\frac{h}{(d - h)} - Z\right)\right) \tanh\left\{\frac{(\Psi_{\rm up} - \Psi_{\rm g})}{4}\right\}}{1 + \exp\left(\kappa (d - h) \left(\frac{h}{(d - h)} - Z\right)\right) \tanh\left\{\frac{(\Psi_{\rm up} - \Psi_{\rm g})}{4}\right\}} \right]$$

$$(13) 256$$

where $\kappa = ((\varepsilon_2 \varepsilon_0 k_B T)/(2e^2 z^2 n_\infty))^{1/2}$ is the inverse of Debye 257 length. The nondimensional interface electric potential, Ψ_s , can 258 be found by solving the following nonlinear equation 259

$$\Psi_{s} - \frac{2\varepsilon_{2}\kappa h}{\varepsilon_{1}} \sinh\left(\frac{(\Psi_{\rm up} - \Psi_{s})}{2}\right) = 0 \tag{14}$$

It is assumed that the IL is a symmetric $(\overline{z}; \overline{z})$ and monovalent 261 electrolyte. It should be noted that use of long wave 262 approximation is valid for the early stages of film evolution. 263 Linear stability (LS) analysis shows that using long wave limit 264 approximation overestimate the maximum wavelength of 265 instability growth when compared to a general model. 266 Using the electric potential distribution obtained from eqs 12 and 267 13 and combining with the electrostatic pressure in eq 6 results in 268 a conjoining pressure acting on the IL—PD bilayer interface of 269

$$\phi_{\rm EL} = -\frac{1}{2}\varepsilon_1 \varepsilon_0 \left(\frac{\varepsilon_1}{\varepsilon_2} - 1\right) \left(\frac{\psi_{\rm s}}{h}\right)^2 \tag{15}$$

For bilayers that act like perfect dielectrics the electrostatic component of conjoining pressure is 29

273

$$\phi_{\rm EL} = -\frac{1}{2}\varepsilon_1 \varepsilon_0 \left(\frac{\varepsilon_1}{\varepsilon_2} - 1\right) \left[\frac{\psi_{\rm up}}{h\left(1 - \frac{\varepsilon_1}{\varepsilon_2}\right) + \frac{\varepsilon_1}{\varepsilon_2}d}\right]^2 \tag{16}$$

Linear Stability Analysis. To predict the characteristic wavelength for growth of instabilities, linear stability analysis has the been performed for PD-PD bilayers 12,16,27,36,37,55,57 and PD-IL bilayers. For a thin PD liquid film bounded with an IL layer, the LS analysis is found by substituting the interface height, h, in eq. 4 the exp $[i(k^*x + k^*y) + st]$ and neglecting the nonlinear terms. k^* is wavenumber, s is the growth coefficient, and ξ is the amplitude. The linear dispersion relation, s, and the resulting dominant wavenumber, k^*_{max} of the fastest growing wave are $s = ([(\partial \phi/\partial h)])[(\partial \phi/\partial h)][(\partial \phi/\partial h)$

Since the bilayer interface potential is a function of interface polarition points applied voltage, electric permittivity of layers, and the Debye length (eq 14), changing any of these parameters changes the interface potential and the electrostatic component of conjoining pressure and consequently the shape and size of the structures. For simulation, time $(T=t/t_s)$ and length $(X=x/\lambda_{\rm max})$ and $(X=y/\lambda_{\rm max})$ are scaled using $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ are scaled using $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ are scaled using $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ are scaled using $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ are scaled using $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ are scaled using $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ are scaled using $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ are scaled using $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ are scaled using $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ are scaled using $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda_{\rm max})$ are scaled using $(X=x/\lambda_{\rm max})$ and $(X=x/\lambda$

Numerical Method. To obtain transient behavior of the 300 liquid—liquid interface, the governing equations (eqs 4 and 5) are 301 solved numerically. Solution of this 4th order nonlinear partial 302 differential equation (PDE) employs an adaptive time step 303 ordinary differential equation (ODE) solver for the temporal and 304 a finite difference method for the spatial direction. First, spatial 305 derivatives are discretized using finite difference to convert the 306 PDE to a differential algebraic equation (DAE) in time. 307 Differential algebraic solver (DASSL) with an adaptive time stepping is then used for the DAE system.⁵⁸ A domain with 309 length of $\lambda_{\rm max}$ or larger is sufficient to see the growth of instabilities when periodic boundary conditions are chosen; otherwise, fluctuations will damp out over time.²⁴ For notation simplicity, λ will be used instead of λ_{max} . A square domain with size of $(4 \times 4)\lambda^2$ and a periodic boundary condition for both x and y directions are chosen. A uniform Cartesian grid is used throughout this study with 121 × 121 spatial grids. Initial 316 conditions for the simulation are a random disturbance while 317 maintaining liquid film volume. The simulation parameters and constants used in this study are listed in Table 1.

■ RESULTS AND DISCUSSION

The use of an external electric field induces instabilities on the deformable film interface to create a variety of structures from simple hexagonal ordered columns, the bicontinuous and holes holes holes holes to the more complex ones such as cage-like, hierarchical, core—shell, and embedded structures from the structures from the simple hexagonal ordered columns, the bicontinuous and the simple holes holes holes holes holes holes for the more complex ones such as cage-like, so hierarchical, core—shell, and embedded structures from the simple hexagonal ordered columns, the bicontinuous and several holes holes

Table 1. Constants or Parameters Used in Simulations

parameter	value
electron charge magnitude (e)	$1.602 \times 10^{-19} \text{ C}$
Boltzmann constant $(k_{\rm B})$	$1.378 \times 10^{-23} \text{ J/K}$
Avogadro number $(N_{\rm A})$	$6.022 \times 10^{23} \text{ 1/mol}$
free space electric permittivity $(arepsilon_0)$	$8.85 \times 10^{-12} \text{ C/V m}$
viscosity of liquid film (μ)	1 Pa s
effective Hamaker constant (A)	$-1.5 \times 10^{-21} \mathrm{J}$
interfacial tension (γ)	0.048 N/m
electric permittivity of the liquid film $(arepsilon_1)$	2.5 (-)
electric permittivity of the bounding media (ε_2)	1-36 (-)
molarity (M)	0.01-1 mmol/L
initial film thickness (h_0)	20-85 nm
electrodes distance (d)	100 nm
Born repulsion cut off distance (l_0)	1-8 nm
applied voltage $(\psi_{ ext{up}})$	0.25-70 V

tures. The film interface structure changes under a homogeneous 325 electric field based on the relative electric permittivity difference 326 of layers which determines the net electric force direction. 327 Referring to eq 16, for the case of $\varepsilon_1 > \varepsilon_2$ ($\varepsilon_1 < \varepsilon_2$) the electrostatic 328 force acts upward (downward) which results in a disjoining 329 (conjoining) pressure.

In ionic liquids, migration and redistribution of free ions result 331 in double layer formation close to the interface. This 332 accumulation of ions alters the electric potential distribution 333 within the IL layer and depends on the ionic strength which is 334 characterized by the Debye length, k^{-1} . The conductivity of ionic 335 liquids increases with high molarity and lower Debye length. The 336 analytical solution, eqs 12 and 13, is examined for three values of 337 molarity in the IL–PD bilayer and results are compared to the 338 electric potential distribution in a PD–PD bilayer in Figure 2a. 339 f2 Electric permittivity ratio, $\varepsilon_r = 0.6$, and applied potential, $\phi_{\rm up} = 340$ 0.25 V, are kept constant.

Ionic liquids with higher molarity, M = 1 mmol/L (lower 342) Debye length $k^{-1} = 1.08$ nm) act like perfect conductors and 343 electric potential decreases slightly near the interface, whereas a 344 potential drop close to the interface is more significant when the 345 ionic strength is lower, M = 0.01 mmol/L (higher Debye length, 346 k^{-1} = 10.8 nm). For the PD film, as expected from solution of the 347 Laplace equation, electric potential decreases linearly near zero at 348 the lower electrode (see Figure 2a). The proposed analytical 349 solution, eqs 13 and 14, gives reasonable results for ionic liquids 350 with $M \ge 0.01$ mmol/L (i.e., $k^{-1} \le 10.8$ nm). The maximum 351 growth rate, s_{max} , and its corresponding maximum wavenumber, 352 k_{max}^* are found for IL-PD bilayers from LS analysis. Growth rates 353 as a function of wavenumber for IL-PD and PD-PD bilayers are 354 compared in Figure 2b. The maximum wavenumber is increased 355 when the PD bounding layer is replaced with the IL one which 356 results in lower maximum wavelength, λ_{\max} , resulting in smaller 357 center to center distance of pillars.10

As shown in Figure 2a, higher ionic strength leads to lower 359 electric potential drop within IL layer. Therefore, the lower layer 360 film experience a higher electric field. Consequently a IL–PD 361 interface is subject to higher electrostatic pressure, $\phi_{\rm EL}$ (eq 15). 362 Effects of electrolyte molarity on the net electrostatic pressure 363 acting on the IL–PD interface for two electric permittivity ratio 364 of layers, ε_r = 0.6 and 2.5, are shown in Figure 3a. In all cases, the 365 f3 magnitude of electrostatic pressure decays as the interface height 366 increases. The negative values for $\phi_{\rm EL}$ represent the upward 367 disjoining pressure and positive values are downward conjoining 368 pressure.

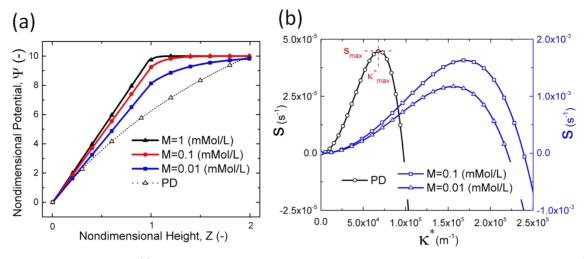


Figure 2. Effects of IL layer molarity on (a) nondimensional electric potential, $\Psi = e\psi/k_{\rm B}T$, variations versus nondimensional height, Z = z/(d-h) and (b) growth rate, s, variations versus wavenumber, k^* . d = 100 nm, $h_0 = 50$ nm, $\psi_{\rm up} = 0.25$ V, $\varepsilon_1 = 2.5$, and $\varepsilon_2 = 4.17$.

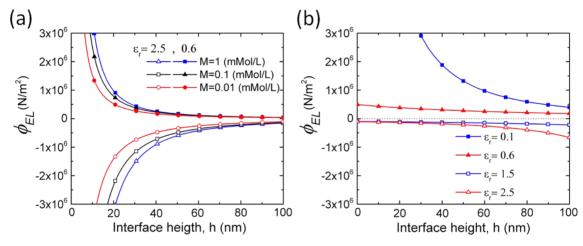


Figure 3. Electrostatic pressure variations versus interface height. Effects of (a) electrolyte molarity in IL-PD bilayers, (b) relative electric permittivity ratios of layers in PD-PD bilayers, ε_r . Applied potential, ψ_{uv} , is (a) 0.25 and (b) 20 V and ε_1 = 2.5.

The effects of relative electric permittivity ratios of the fluid 371 layers in PD-PD bilayers on the electrostatic conjoining/ disjoining pressure is presented in Figure 3b. In this case the applied potential, $\psi_{\rm up}$ = 20 V, is 80 times greater than that used 373 for the IL–PD bilayers (Figure 3a). Comparing $\phi_{\rm EL}$ values for ε_r 374 0.6 and 2.5 between IL-PD bilayers and PD-PD bilayers show that electrostatic pressure varies more uniformly with 376 change of interface height in PD-PD bilayers. Using bounding layers with high relative electric permittivity in PD-PD bilayers also leads to higher electrostatic pressure with similar exponential 380 decrease as a function of interface height to IL-PD bilayers. 381 Although the applied potential is 80 times smaller in the IL-PD 382 case, the interface experience higher electric pressure even without considering the change in molarity. 383

When the spinodal parameter (derivation of conjoining 385 pressure with respect to the film thickness) becomes negative, 386 the film becomes unstable. For the PD–PD bilayers (eq 16), 387 both cases $\varepsilon_1 > \varepsilon_2$ and $\varepsilon_1 < \varepsilon_2$ result in negative spinodal 388 parameter values. However, for the IL–PD bilayers (eq 15), the 389 spinodal parameter is negative only for bilayers with $\varepsilon_1 < \varepsilon_2$. 390 Therefore, this study focuses on the unstable case, where the 391 upper layer has a higher electric permittivity than the lower layer 392 ($\varepsilon_1 < \varepsilon_2$).

Variations in electrostatic pressure due to change in applied $_{393}$ potential for PD–PD and IL–PD bilayers are presented in $_{394}$ Figure 4. To show the broad range of variations in $\phi_{\rm EL}$ results are $_{395~f4}$ plotted on a log–log scale. An electric permittivity ratio of ε_r = $_{396}$ 0.6 and the IL layer molarity of M = 0.1 mmol/L (k^{-1} = 3.43 nm) $_{397}$ are constant. The IL–PD interfaces is subject to a much higher $_{398}$

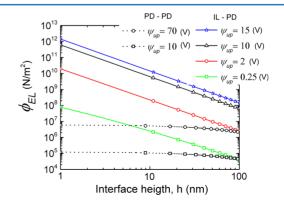


Figure 4. Electrostatic pressure variations versus interface height in PD-PD bilayer and IL-PD bilayer, considering applied potential effects. M = 0.1 mmol/L for IL-PD, $\varepsilon_1 = 2.5$ and $\varepsilon_2 = 4.17$.

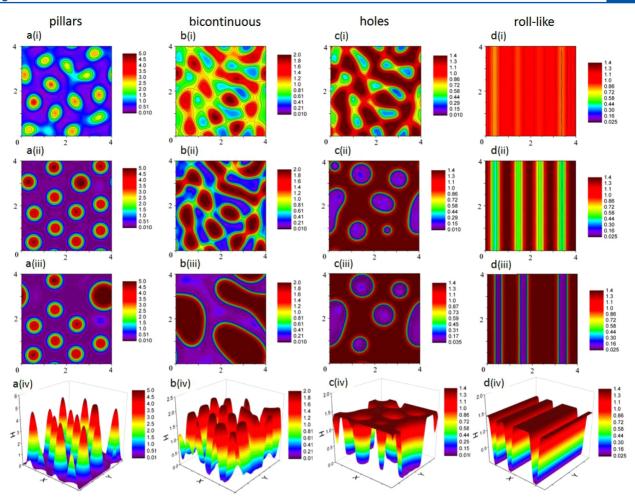


Figure 5. Base case PD-PD bilayer (a-d) images from i to iii show the 2-D spatiotemporal evolution for liquid—liquid interface instabilities in a $(4 \times 4)\lambda^2$ domain when $\varepsilon_r = 0.6$ and $\varepsilon_1 = 2.5$. Initial mean film thicknesses (h_0) are a(i-iv) 20 nm, b(i-iv) 50 nm, c(i-iv) 70 nm, and d(i-iv) 85 nm. Initial electric field intensities (E_0) are 294, 250, 227, and 212 MV/m, respectively. Nondimensional times for the 2-D plots are $T = a(i) 9 \times 10^4$, a(ii) 3.4×10^5 , and a(iii) 2.4×10^6 ; b(i) 8×10^2 , b(ii) 1×10^3 , and b(iii) 4×10^3 ; c(i) 2×10^2 , c(ii) 6×10^2 , and c(iii) 9×10^2 ; d(i) 1.8×10^3 , d(ii) 2.2×10^3 , and d(iii) 6.1×10^3 . Plots a—d(iv) show 3-D snapshots of the liquid—liquid interface structure. Nondimensional times for the plots are $T = a(iv) 3.4 \times 10^5$, b(iv) 1×10^3 , c(iv) 6×10^2 , and d(iv) 6.1×10^3 .

electrostatic pressure compared with the PD-PD interface (in the order of 10^3 to 10^7 times higher for the $\psi_{up} = 10 \text{ V}$ case).

401 IL—PD bilayers are unstable under applied electric field while 402 $\varepsilon_r < 1$. To investigate this response, PD—PD bilayers with $\varepsilon_r < 1$ 403 are first considered by calculating the dynamics and resulting 404 spatiotemporal evolution of interface. This provides a basis of 405 comparison for unstable IL—PD bilayers in the next step.

Perfect Dielectric—Perfect Dielectric Bilayers (ε_r < 1). 407 Bilayers with ε_r < 1, which act like perfect dielectric materials (i.e., 408 there is no free charge or free ion in media) are examined first as a 409 baseline for comparison to IL—PD bilayers. Electric breakdown 410 in layers is not included in the simulations and it is assumed that 411 the film layers behave like perfect dielectrics during the 412 patterning process. The applied voltage, $\psi_{\rm up}$, is set to a low 413 value (20 V), but even at this low applied voltage, the expected 414 electric field within the layers may exceed the dielectric field 415 strength of the material and result in electric breakdown violating 416 the model assumption. ³⁹

417 A 2-D spatiotemporal evolution and 3-D snapshots of the 418 liquid—liquid interface instabilities for the homogeneous electric 419 field and PD—PD bilayer with ε_r = 0.6 is shown in Figure 5. Initial 420 mean film thicknesses (h_0) are a(i-iv) 20 nm, b(i-iv) 50 nm, c(i-421 iv) 70 nm, and d(i-iv) 85 nm. The electrostatic conjoining

pressure pushes the interface downward, so some hollows in the 422 lower layer are expected to form. Negative diffusion 29 moves the 423 fluid from the thinner to the thicker region which assists in the 424 growth of instabilities over time. Four distinct structures of the 425 following: pillars a(iv), bicontinuous b(iv), holes c(iv), and roll- 426 like d(iv) are observed. Similar to the bilayers with $\varepsilon_r > 1$, 29,41 427 pillars and bicontinuous structures form in PD–PD systems. 428 However, holes and roll-like structures only form when there is a 429 conjoining pressure for $\varepsilon_r < 1$. 27,30

Pillars which have a hexagonal structure a(ii) imply that 431 coalescence of the neighboring pillars a(iii) is due to their 432 collision. S5 Bicontinuous structures, b(ii) and b(iii), behave like 433 an air-in-liquid dispersion which occurs in air-polymer (i.e., 434 monolayer) system with high filling ratios. For hole formation 435 the interface touches the upper electrode in the early stages of 436 process, c(i), with dumbbell shape dents on the interface. These 437 dents evolve into round shapes with time and deepen toward the 438 lower electrode, c(ii). Compared to the pillars, a(ii), the holes, 439 c(ii), do not have a hexagonal order over the domain; therefore, 440 Ostwald ripening is one possible reason for the coalescence of 441 neighbor holes over time. Roll-like or straight groove-type 442 structures seem to be an organized form of the bicontinuous 443 structures generated by the same phase inversion mechanism. 444

445 Images a—d(ii) show the quasi-equilibrium structures and a—446 d(iii) are snapshots for the coalescence stage. The time that 447 patterns remain in the quasi-equilibrium condition depends on 448 physical parameters of the system like viscosity, electric 449 permittivity ratio of layers, and initial mean thickness. ^{29,41,55}

To characterize different morphological behaviors in bilayers 451 with $\varepsilon_r = 0.6$ and different thicknesses, as shown in Figure 5, a 452 systematic numerical investigation is performed. A parametric 453 map that captures the liquid—liquid interface structure for wide 454 ranges of relative electric permittivity ratios of layers, ε_r , and 455 nondimensional initial mean film thickness, h_0/d , is shown in 456 Figure 6. Both initial layer thickness and electric permittivity ratio

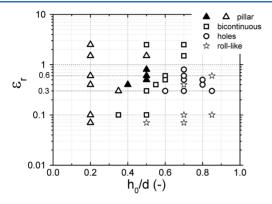


Figure 6. 2-D parametric map that illustrates the liquid—liquid interface structure as a function relative electric permittivity ratios of layers, ε_r , and nondimensional initial mean film thickness, h_0/d .

are critical factors in the EHL process and can create different 457 patterns on the film. When the lower layer is thicker than the 458 upper layer, the final pattern is more sensitive to electric 459 permittivity. Changing the electric permittivity ratio of the 460 bilayers results in the interfaces having two to four different kinds 461 of structure depending on the initial thicknesses of the layers. For 462 example, in the case of $\varepsilon_r = 1.5$, pillars and bicontinuous 463 structures will form when $h_0/d = 0.2$ and 0.5, respectively; but for 464 $\varepsilon_r = 0.3$, pillars, bicontinuous, and hole structures are observed 465 for h_0/d ratios of 0.35, 0.5, and 0.6, respectively. Moreover, the 466 interface can have four distinct structures, namely, pillars, 467 bicontinuous, hole, and roll-like where $\varepsilon_r = 0.6$ and h_0/d ratios 468 of 0.2, 0.6, 0.7, and 0.85, respectively.

From a thermodynamic point of view, the final structure in the 470 pattern formation process is the shape that is at the lowest 471 Helmholtz free energy of the system which is in a thermodynami- 472 cally stable state condition. In Figure 6, a critical initial film 473 thickness for each ε_r , can be found such that the EHD patterning 474 process of films with initial thicknesses below this critical value 475 results in pillar formation. The critical film thickness is called the 476 Maxwell point 17,30 which is the inflection point in the plot of 477 conjoining/disjoining pressure versus interface height in the 478 steady state solution. Warying the relative electric permittivity 479 ratio of the layers changes the conjoining/disjoining pressure 480 (see Figure 3) which results in a shift to the critical film thickness 481 value.

In some cases, structures do not last for a long time, 483 particularly for patterns that are close to the boundaries between 484 different structures in Figure 6. Filled triangles in Figure 6 denote 485 the unstable pillars which coalesce quickly and form 486 bicontinuous structures. These are bounded between stable 487

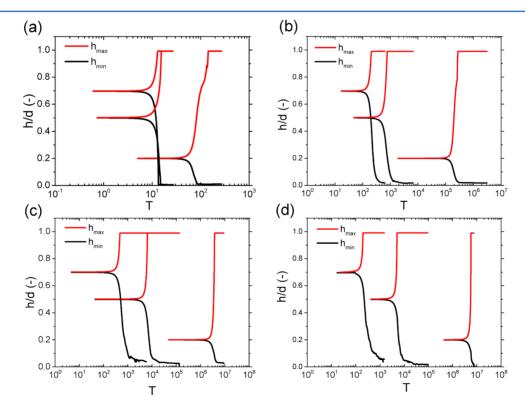


Figure 7. Nondimensional maximum and minimum film thickness, h/d (-), variations versus nondimensional time, T. Relative electric permittivity ratios of layers, ε_r , and initial mean film thickness, h_0 , effects on the drainage time. Permittivity ratios are $\varepsilon_r = (a) 0.1$, (b) 0.6, (c) 1.5, and (d) 2.5. Initial electric field intensities (E_0) for 20-, 50-, and 70-nm-thick films are (a) 714, 363, and 273 MV/m; (b) 294, 250, and 227 MV/m; (c) 142, 160, and 174 MV/m; and (d) 90, 114, and 137 MV/m, respectively.

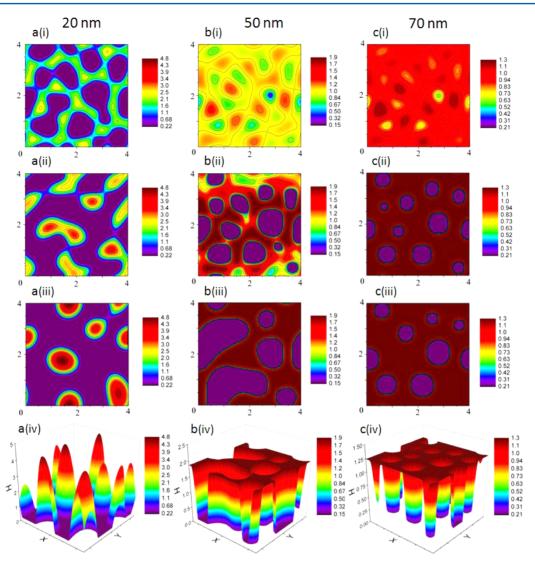


Figure 8. IL—PD bilayer (a—c) images from i to iii show the 2-D spatiotemporal evolution for liquid—liquid interface instabilities in a $(4 \times 4)\lambda^2$ domain when $\varepsilon_r = 0.6$ and M = 100 (mmol/L). Initial mean film thicknesses (h_0) are a(i-iv) 20 nm, b(i-iv) 50 nm, and c(i-iv) 70 nm. Nondimensional times for the plots are T = a(i) 9, a(ii) 11, and a(iii) 13.5; b(i) 7, b(ii) 8, and b(iii) 20; c(i) 7, c(ii) 10, and c(iii) 17. Plots a—c(iv) show 3-D snapshots of the liquid—liquid interface structure. Nondimensional times for the 3-D plots are T = a(iv) 13.5, b(iv) 20, and c(iv) 10. Initial electric field intensities (E_0) are (a) 90, (b) 11, and (c) 137 MV/m.

488 pillars and bicontinuous structures. By careful examination of 489 Figure 6, it is found that as the mean initial thickness of the layers 490 is varied (h_0/d) , more types of structures are produced when ε_r < 491 1 is compared to $\varepsilon_r > 1$.

The effect of relative electric permittivity ratios of layers, ε_n on 493 the nondimensional maximum and minimum film thicknesses, 494 h_{max}/d and h_{min}/d , over nondimensional time, T, for three polymer film thicknesses of 20, 50, and 70 nm is shown in Figure 496 7. Four values of (a) 0.1, (b) 0.6, (c) 1.5, and (d) 2.5 are chosen 497 for the ε_r to compare the bilayer behavior for $\varepsilon_r > 1$ and $\varepsilon_r < 1$. The parameter h_{\min} indicates when the lower layer drains (or 499 nonwets the lower electrode) and $h_{\rm max}$ indicates when the lower layer touches the upper electrode in the domain. To compare bilayers with different thicknesses, interface height, h, is 502 nondimensionalized using electrode distance as a length scale. The actual time in real experiments depends on the properties of 504 the applied materials, mainly viscosity. For bilayers with $\varepsilon_r < 1$, 505 thicker films drain more rapidly than thinner ones as shown in 506 Figure 7a,b, a similar trend is found for the bilayers with $\varepsilon_r > 1$ 507 shown in Figure 7c,d.

lonic Liquid—Perfect Dielectric Bilayers ($\varepsilon_{\rm r}$ < 1). When 508 IL—PD bilayers are considered, it was found that the interface 509 experiences higher electric field. The electrostatic model and 510 corresponding electrostatic pressure derived earlier are used in 511 the thin film equation to examine the spatiotemporal evolution of 512 IL—PD interface. The electric permittivity ratio of bilayer is set to 513 $\varepsilon_{\rm r}=0.6$, and the other physical parameters are kept the same as 514 those of PD—PD bilayers for comparison purpose. Three bilayers 515 with initial mean lower layer thicknesses (h_0) of 20, 50, and 70 516 nm are simulated and the 2-D spatiotemporal evolution and 3-D 517 snapshots are shown in Figure 8.

For the IL–PD cases in Figure 8, a homogeneous electric field 519 is applied in a $(4 \times 4)\lambda^2$ domain at $\varepsilon_r = 0.6$ and M = 100 mM/L. 520 In the IL–PD case with a 20 nm lower layer initial thickness 521 (Figure 8a), pillars form on the interface similar to the PD–PD 522 system (Figure 5a). However, the generated pillars are more 523 sparsely spread and have an elliptical cross section instead of 524 circular. Increasing h_0 to 50 nm causes the IL–PD to switch to 525 polydisperse holes (Figure 8b(ii). In the PD–PD bilayer a 526 bicontinuous structure (Figure 5b(ii)) is formed at $h_0 = 50$ nm, 527

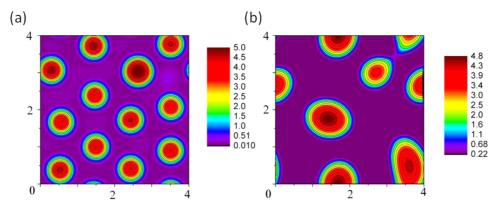


Figure 9. 2-D spatiotemporal evolution for liquid–liquid interface instabilities for the homogeneous electric field in (a) PD–PD and (b) IL–PD bilayers, in a $(4 \times 4)\lambda^2$ domain when $\varepsilon_r = 0.6$ and M = 100 (mmol/L) for the IL–PD bilayer. The initial mean film thickness is $h_0 = 20$ nm and the nondimensional times for the plots are $T = (a) 3.4 \times 10^5$ and (b) 13.5. $\psi_{up} = 20$ V and initial electric field intensities (E_0) are (a) 294 and (b) 995.

528 which is unlike the similar condition IL-PD bilayer holes (Figure 529 8b(ii)). A further increase in h_0 to 70 nm in the IL-PD case leads 530 to the generation of more organized holes (Figure 8c(ii)). Thus, 531 using an ionic liquid in place of a perfect dielectric seems to 532 effectively causes a shift to the left in the boundaries between 533 structure types in Figure 6.

The disordered and polydispersed pillars of the IL—PD system which are captured in the numerical simulation are also experimentally observed. Thus, the numerical simulation developed here in predicting the interface behavior in the IL—538 PD system seems to capture the dynamics of interest. A 2-D system spatiotemporal evolution for liquid—liquid interface instabilities for the homogeneous electric field in (a) PD—PD and (b) IL—541 PD bilayers are directly compared in Figure 9. The number of pillars in the IL—PD bilayer is more than what is immediately apparent in Figure 9, since the physical domain size in IL—PD is to times less than that in PD—PD. Specifically, the average center-to-center distance of pillars is 210 nm for the IL—PD bilayer, duantifying the smaller size features in the IL—PD bilayers for this case.

49 CONCLUSIONS

550 Electrostatic force is used as a destabilizing external force to study 551 the dynamics, morphological evolution, and drainage time of thin 552 liquid bilayers (<100 nm). Due to the geometrical properties of 553 the system (small aspect ratio of mean film thickness to lateral 554 domain) a long-wave asymptotic theory is employed to simplify 555 the coupled mass and momentum equations. This results in a 556 nonlinear thin film equation approximation which is numerically 557 solved. The generated thin film fluid structures of IL–PD 558 bilayers for varying parameters are simulated and compared to 559 the well-known PD–PD base case.

In PD—PD bilayers with $\varepsilon_r < 1$, pillar, bicontinuous, hole, and 561 roll-like structures are formed, whereas the last two features are 562 not observed in the bilayers with $\varepsilon_r > 1$. When the thin film is 563 thicker than bounding media, the final pattern is more sensitive 564 to the electric permittivity ratio. Thicker films are found to drain 565 more rapidly in bilayers with $\varepsilon_r < 1$ similar to bilayers with $\varepsilon_r > 1$. For IL—PD bilayers, to predict the electrostatic component of 567 conjoining/disjoining pressure acting on the interface, an 568 analytical model is developed based on the nonlinear Poission-569 Boltzmann equation. Perfect dielectrics and perfect conductors 570 are found as limiting cases for ionic liquids with low and high ionic strength, respectively. The IL—PD interface is subject to a

much higher electrostatic pressure compared with the PD–PD 572 interface (on the order of 10^3 to 10^7 times higher for the $\psi_{\rm up} = 10$ 573 V case). It is found that the IL–PD bilayers with $\varepsilon_r > 1$ remain 574 stable under the action of transverse electric field, whereas similar 575 PD–PD bilayers are unstable.

The use of an IL layer is found to reduce the size of the structures by a factor of ten ($\lambda_{\rm PD-PD}$) is about ten times larger than 578 $\lambda_{\rm IL-PD}$) for the case studied. This has important implications 579 when the goal is to produce small structures. The IL-PD pillars 580 tend to be polydispersed and more disordered over the domain 581 compared to the PD-PD ones. The developed analytical models 582 and resulting numerical simulation closely predict the interface 583 behavior for both PD-PD and IL-PD bilayers over the range of 584 parameters tested when compared to other numerical studies 585 and experimental observations in the literature. This indicates 586 that the numerical simulation developed is of utility in predicting 587 PD-PD and IL-PD bilayer responses to an applied electric field. 588

AUTHOR INFORMATION

Corresponding Author

*E-mail: bob.koch@ualberta.ca.

Notes

589

590

591

592

600

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

We acknowledge financial support from the Natural Sciences and 595 Engineering Research Council (NSERC), Canada, Chair in 596 Water Quality Management for Oil Sands Extraction, the Alberta 597 Water Research Institute (AWRI), Kemira, Outotec, Suncor 598 Energy, Statoil and ConocoPhilips.

REFERENCES

I

- (1) Quere, D. Fluid Coating on a Fiber. *Annu. Rev. Fluid Mech.* **1999**, 601 31, 347–384.
- (2) Eow, J. S.; Ghadiri, M.; Sharif, A. O.; Williams, T. J. Electrostatic 603 Enhancement of Coalescence of Water Droplets in Oil: A Review of the 604 Current Understanding. *Chem. Eng. J.* **2001**, *84*, 173–192.
- (3) Mostowfi, F.; Khristov, K.; Czarnecki, J.; Masliyah, J.; 606 Bhattacharjee, S. Electric Field Mediated Breakdown of Thin Liquid 607 Films Separating Microscopic Emulsion Droplets. *Appl. Phys. Lett.* **2007**, 608 90, 184102.
- (4) Lin, S.; Brenner, H. Marangoni Convection in a Tear Film. *J.* 610 Colloid Interface Sci. 1982, 85, 59–65.
- (5) Vrij, A. Possible Mechanism for the Spontaneous Rupture of Thin 612 Free Liquid Films. *Discuss. Faraday Soc.* **1966**, 42, 23–33.

f9

- 614 (6) Sheludko, A. Thin Liquid Films. Adv. Colloid Interface Sci. 1967, 1, 615 391–464.
- 616 (7) Ruckenstein, E.; Jain, R. K. Spontaneous Rupture of Thin Liquid 617 Films. J. Chem. Soc., Faraday Trans. 1974, 70, 132–147.
- 618 (8) Williams, M. B.; Davis, S. H. Nonlinear Theory of Film Rupture. *J. Colloid Interface Sci.* **1982**, *90*, 220–228.
- 620 (9) Sharma, A. Relation of Thin Film Stability and Morphology to 621 Macroscopic Parameters of Wetting in the Apolar and Polar Systems. 622 *Langmuir* **1993**, *9*, 861–869.
- 623 (10) Oron, A.; Davis, S. H.; Bankoff, S. G. Long-Scale Evolution of 624 Thin Liquid Films. *Rev. Mod. Phys.* **1997**, *69*, 931–980.
- 625 (11) A. Ghatak, R. K.; Sharma, A. Dynamics and Morphology of Holes 626 in Dewetting of Thin Films. *J. Colloid Interface Sci.* **1999**, 212, 483–494.
- 627 (12) Sharma, A.; Khanna, R. Pattern Formation in Unstable Thin 628 Liquid Films Under the Influence of Antagonistic Short- and Long-629 range Forces. J. Chem. Phys. 1999, 110, 4929—4936.
- 630 (13) Chan, D. Y. C.; Klaseboer, E.; Manica, R. Film Drainage and 631 Coalescence Between Deformable Drops and Bubbles. *Soft Matter* **2011**,
- 633 (14) Sharma, A.; Ruckenstein, E. An Analytical Nonlinear Theory of 634 Thin Film Rupture and Its Application to Wetting Films. *J. Colloid* 635 *Interface Sci.* 1986, 113, 456–479.
- 636 (15) Mitlin, V. S. Dewetting of Solid Surface: Analogy with Spinodal 637 Decomposition. *J. Colloid Interface Sci.* **1993**, *156*, 491–497.
- 638 (16) Schaffer, E.; Thurn-Albrecht, T.; Russell, T. P.; Steiner, U. 639 Electrically Induced Structure Formation and Pattern Transfer. *Nature* 640 **2000**, 403, 874–877.
- 641 (17) Bestehorn, M.; Pototsky, A.; Thiele, U. 3D Large Scale Marangoni 642 Convection in Liquid Films. *Eur. Phys. J. B* **2003**, 33, 457–467.
- 643 (18) Schaffer, E.; Harkema, S.; Roerdink, M.; Blossey, R.; Steiner, U. 644 Morphological Instability of a Confined Polymer Film in a Thermal 645 Gradients. *Macromolecules* **2003**, *36*, 1645–1655.
- (19) Wang, Q.; Zhang, L.; Zhao, X. Creasing to Cratering Instability in
 Polymers under Ultrahigh Electric Fields. *Phys. Rev. Lett.* 2011, 106,
 118301.
- 649 (20) Corbett, A.; Kumar, S. Combined Thermal and Electro-650 hydrodynamic Patterning of Thin Liquid Films. *J. Eng. Math.* **2013**, 651 1–16.
- 652 (21) Trice, J.; Favazza, C.; Thomas, D.; Garcia, H.; Kalyanaraman, R.; 653 Sureshkumar, R. Novel Self-Organization Mechanism in Ultrathin 654 Liquid Films: Theory and Experiment. *Phys. Rev. Lett.* **2008**, *101*, 655 017802.
- 656 (22) Krishna, H.; Shirato, N.; Yadavali, S.; Sachan, R.; Strader, J.; 657 Kalyanaraman, R. Self-Organization of Nanoscale Multilayer Liquid 658 Metal Films: Experiment and Theory. *ACS Nano* **2011**, *5*, 470–476.
- 659 (23) Berendsen, C. W. J.; Zeegers, J. C. H.; Darhuber, A. A. Thinning 660 and Rupture of Liquid Films by Moving Slot Jets. *Langmuir* **2013**, 29, 661 15851–15858.
- 662 (24) Khanna, R.; Sharma, A. Pattern Formation in Spontaneous 663 Dewetting of Thin Apolar Films. *J. Colloid Interface Sci.* **1997**, *195*, 42–664 50.
- 665 (25) Israelachvili, J. N. *Intermolecular and Surface Forces*; Academic 666 Press: Burlington, MA. 2011.
- 667 (26) Masliyah, J. H.; Bhattacharjee, S. Electrokinetic and Colloid 668 Transport Phenomena; Wiley-Interscience: Hoboken, NJ, 2006.
- 669 (27) Lin, Z.; Kerle, T.; Russell, T. P.; Schäffer, E.; Steiner, U. Structure 670 Formation at the Interface of Liquid/Liquid Bilayer in Electric Field. 671 *Macromolecules* **2002**, *35*, 3971–3976.
- 672 (28) Salac, D.; Wei, L.; Chia-Wei, W.; Sastry, A. M. Pattern Formation 673 in a Polymer Thin Film Induced by an In-plane Electric Field. *Appl. Phys.* 674 Lett. **2004**, 85, 1161–1163.
- 675 (29) Verma, R.; Sharma, A.; Kargupta, K.; Bhaumik, J. Electric Field 676 Induced Instability and Pattern Formation in Thin Liquid Films. 677 *Langmuir* **2005**, *21*, 3710–3721.
- 678 (30) Wu, N.; Russel, W. B. Electrohydrodynamic Instability of 679 Dielectric Bilayers: Kinetics and Thermodynamics. *Ind. Eng. Chem. Res.* 680 **2006**, 45, 5455–5465.

- (31) Kim, D.; Lu, W. Three-Dimensional Model of Electrostatically 681 Induced Pattern Formation in Thin Polymer Films. *Phys. Rev. B* **2006**, 682 73, 35206.
- (32) Wu, N.; Russel, W. B. Micro- and Nano-patterns Created via 684 Electrohydrodynamic Instabilities. *Nano Today* **2009**, *4*, 180–192. 685
- (33) Voicu, N. E.; Harkema, S.; Steiner, U. Electric-Field-Induced 686 Pattern Morphologies in Thin Liquid Films. *Adv. Funct. Mater.* **2006**, *16*, 687 926–934.
- (34) Verma, R.; Sharma, A.; Banerjee, I.; Kargupta, K. Spinodal 689 Instability and Pattern Formation in Thin Liquid Films Confined 690 Between Two Plates. *J. Colloid Interface Sci.* **2006**, 296, 220–232.
- (35) Bae, J.; Glogowski, E.; Gupta, S.; Chen, W.; Emrick, T.; Russell, T. 692 P. Effect of Nanoparticles on the Electrohydrodynamic Instabilities of 693 Polymer/Nanoparticle Thin Films. *Macromolecules* **2008**, 41, 2722–694 2726.
- (36) Roberts, S. A.; Kumar, S. AC Electrohydrodynamic Instabilities in 696 Thin Liquid Films. *J. Fluid Mech.* **2009**, *631*, 255–279.
- (37) Roberts, S. A.; Kumar, S. Electrohydrodynamic Instabilities in 698 Thin Liquid Trilayer Films. *Phys. Fluids* **2010**, 22, 122102. 699
- (38) Reddy, P. D. S.; Bandyopadhyay, D.; Sharma, A. Self-Organized 700 Ordered Arrays of Core-Shell Columns in Viscous Bilayers Formed by 701 Spatially Varying Electric Fields. *J. Phys. Chem. C* **2010**, *114*, 21020 702 21028
- (39) Lau, C. Y.; Russel, W. B. Fundamental Limitations on Ordered 704 Electrohydrodynamic Patterning. *Macromolecules* **2011**, *44*, 7746–7751. 705
- (40) Gambhire, P.; Thaokar, R. Linear Stability Analysis of 706 Electrohydrodynamic Instabilities at Fluid Interfaces in the Small 707 Feature Limit. *Eur. Phys. J. E* **2011**, *34*, 1–12.
- (41) Atta, A.; Crawford, D. G.; Koch, C. R.; Bhattacharjee, S. Influence 709 of Electrostatic and Chemical Heterogeneity on the Electric-Field-710 Induced Destabilization of Thin Liquid Films. *Langmuir* **2011**, 27, 711 12472–12485.
- (42) Gambhire, P.; Thaokar, R. M. Role of Conductivity in the 713 Electrohydrodynamic Patterning of Air-Liquid Interfaces. *Phys. Rev. E* 714 **2012**. 86. 036301.
- (43) Wang, Q.; Zhao, X. Creasing-Wrinkling Transition in Elastomer 716 Films under Electric Fields. *Phys. Rev. E* **2013**, *88*, 042403.
- (44) Mondal, K.; Kumar, P.; Bandyopadhyay, D. Electric Field Induced 718 Instabilities of Thin Leaky Bilayers: Pathways to Unique Morphologies 719 and Miniaturization. *J. Chem. Phys.* **2013**, *138*, 024705.
- (45) Roy, S.; Biswas, D.; Salunke, N.; Das, A.; Vutukuri, P.; Singh, R.; 721 Mukherjee, R. Control of Morphology in Pattern Directed Dewetting of 722 a Thin Polymer Bilayer. *Macromolecules* **2013**, *46*, 935–948.
- (46) Benney, D. J. Long Waves on Liquid Films. J. Math. Phys. 1966, 724 45, 150–155.
- (47) Yang, Q.; Li, B. Q.; Ding, Y. Dynamic Modelling of Micro/Nano-726 patterning Transfer by an Electric Field. *RSC Adv.* **2013**, 3, 24658–727 24663.
- (48) Yang, Q.; Li, B. Q.; Ding, Y. A Numerical Study of Nanoscale 729 Electrohydrodynamic Patterning in a Liquid Film. *Soft Matter* **2013**, *9*, 730 3412–3423.
- (49) Saville, D. A. Electrohydrodynamics: The Taylor-Melcher Leaky 732 Dielectric Model. *Annu. Rev. Fluid Mech.* **1997**, 27–64.
- (50) Craster, R. V.; Matar, O. K. Electrically Induced Pattern 734 Formation in Thin Leaky Dielectric Films. *Physics of Fluids* (1994-735 present) **2005**, 17, 1131–1198.
- (51) Gambhire, P.; Thaokar, R. Electrokinetic Model for Electric-737 Field-Induced Interfacial Instabilities. *Phys. Rev. E* **2014**, *89*, 032409.
- (52) Lee, D. W.; Im, D. J.; Kang, I. S. Electric Double Layer at the 739 Interface of Ionic Liquid-Dielectric Liquid under Electric Field. 740 Langmuir 2013, 29, 1875—1884.
- (53) Atherton, R. W.; Homsy, G. M. On the Derivation of Evolution 742 Equations for Interfacial Waves. *Chem. Eng. Commun.* **1976**, 2, 57–77. 743
- (54) Sharma, A.; Khanna, R. Pattern Formation in Unstable Thin 744 Liquid Films. *Phys. Rev. Lett.* **1998**, *81* (16), 3463–3466.
- (55) Wu, N.; Kavousanakis, M. E.; Russel, W. B. Coarsening in the 746 Electrohydrodynamic Patterning of Thin Polymer Films. *Phys. Rev. E* 747 **2010**, *81*, 26306.

- 749 (56) Landau, L. D.; Lifshitz, E. M. Electrodynamics of continuous media; 750 Pergamon Press, 1960.
- 751 (57) Bandyopadhyay, D.; Sharma, A.; Thiele, U.; Reddy, P. D. S.
- 752 Electric-Field-Induced Interfacial Instabilities and Morphologies of
- 753 Thin Viscous and Elastic Bilayers. *Langmuir* **2009**, 25, 9108–9118.
- 754 (58) Brenan, K. E.; Petzold, L. R. The Numerical Solution of Higher
- 755 Index Differential/Algebraic Equations by Implicit Methods. SIAM J.
- 756 Numer. Anal. **1989**, 26 (4), 976–996.
- 757 (59) Dickey, M. D.; Gupta, S.; Leach, K. A.; Collister, E.; Willson, C.
- 758 G.; Russell, T. P. Novel 3-D Structures in Polymer Films by Coupling
- 759 External and Internal Fields. Langmuir 2006, 22, 4315-4318.
- 760 (60) Morariu, M. D.; Voicu, N. E.; Schaffer, E.; Lin, Z.; Russell, T. P.;
- 761 Steiner, U. Hierarchical Structure Formation and Pattern Replication
- 762 Induced by an Electric Field. Nat. Mater. 2003, 2, 48-52.